

A Comparative Study of Synthesis and Characterization of Pure and Dual Doped (Cu-Co) α -MnO₂ Nanorods

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Abstract:

This study synthesized manganese oxide (α -MnO₂) nanoparticles with dual doping of cobalt (Co) and copper (Cu) nanorods. We investigated optimal conditions in Cu-Co nanorods doped with α -MnO₂. The successful incorporation of cobalt and copper was anticipated using X-ray diffraction. The average particle size of dual doped α -MnO₂ nanoparticles was estimated using XRD analysis. The functional group analysis was evaluated using FTIR Spectroscopy.

This co-precipitation approach provides advantages such as a simple and speedy preparative method, as well as easy control of particle size and composition, making it commercially frequently utilized due to its cost effectiveness. MnO₂ is one of the most popular catalysis materials because of its unique qualities such as high activity, low cost, low toxicity, and environmental compatibility.

Keywords: Dual doped α -MnO₂, XRD, Nanorods

Introduction:

Nanotechnology is the study and technology of materials with at least one of three dimensions less than one hundred nanometers. [1] Particles are important in study because of their distinct optical, structural, and magnetic properties. Several factors determine nanoparticle properties, including shape, size, composition, and structure [2]. Nanoparticle size is determined by their dimension and form, resulting in materials with consistent properties. Nanoparticle formation is dependent on a regulated core structure and size [3, 4]. Metal nanoparticles having a significant specific surface area have been extensively studied due to their distinct physical and electrical properties [5]. Manganese oxide nanoparticles have good physicochemical properties and can be employed in a wide range of applications, including catalysts, molecular sieves, batteries, magnetic materials, and more. The optical and electrical properties of produced manganese oxide nanoparticles are important aspects to consider for future research in solar cells [6].

Transition metal oxide nanostructures have emerged as a viable electrode material for energy storage [7]. Manganese is a transition element with three different valence states, and its oxides are regarded

very complicated [8]. Among the transition metal oxides, manganese exhibits multiple oxidation states and hence forms distinct oxides (MnO , Mn_2O_3 , Mn_3O_4 , Mn_5O_8 , and MnO_2) [9]. Nanoparticles, due to their large surface area, can perform both Faradic and Non-Faradaic charge transfer mechanisms. Kumar et al. and Balamurugan generated nanocrystalline manganese oxide nanoparticles with tetragonal structure via co-precipitation with two distinct anions salts (sulphate monohydrate and oxalate) [10, 11]. Wu et al. found that hydrothermal synthesis could generate many morphologies of MnO_2 nanostructures, including α - MnO_2 nanorods, nanotubes, and nanowires [12]. In this regard, manganese oxide (MnO_2) with its tetragonal structure and (2x2) tunnelled type Hollandite (alpha) network makes it a promising option for supercapacitor applications [13]. α - MnO_2 is most relevant due to its great chemical stability and long cycle life when used in electrochemical capacitors. [14].

Co oxidation was also observed on copper and manganese oxide, with the presence of Cu^{2+} and Mn^{3+} [15] Copper doped MnO_2 is the insertion of copper ions (Cu^{2+}) into the crystal structure of MnO_2 . This doping procedure involves replacing some of the Mn atoms. With Cu atoms, changing the composition and characteristics of MnO_2 . Copper doping can improve total conductivity by enabling charge transport inside the crystal lattice [16]. Cobalt is thought to be one of the most promising metal cations. Cobalt was discovered to be one of the most promising metal cations. Incorporating Co ions into MnO_2 produces a greater pseudo capacitance of cobalt oxide. Co-doped MnO_2 nanoparticles show a significant increase in electrode conductivity, and studies based on Co- MnO_2 suggest that it could be a viable electrode material for several high capacitive applications. [17] MnO_2 has various structural forms, including α , β , γ , δ , ϵ , and λ . The fundamental structural unit (MnO_6 Octahedron) is linked in various ways. [18]

MnO_2 nanoparticles can be prepared using either top-down or bottom-up processes. The top-down technique is not commonly used due to the high preparation cost and structural defects in the generated nanoparticles [19]. Most researchers prefer the bottom-up technique, which produces particles of homogeneous size and morphology. The wet chemical method is utilized to synthesize MnO_2 nanoparticles. This study focuses on summarizing the most commonly used wet chemical procedures, including hydrothermal [20], redox process [21], sol gel method [22], thermal reflux process [23], chemical precipitation method [24], and green synthesis method [25].

2. Experimental part

2.1. Chemical

The chemicals Manganese chloride tetrahydrate ($\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$), copper nitrate trihydrate ($\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$), cobalt nitrate hexahydrate ($\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$) and sodium hydroxide (NaOH) were used the raw materials.

2.2. Synthesis of pure and dual doped α - MnO_2 nanorods

For the synthesis of dual doped α - MnO_2 nanorods, 0.98 g of $\text{MnCl}_2 \cdot 4\text{H}_2\text{O}$ was dissolved in a beaker containing 50 ml of distilled water under ambient temperature. Then, 0.14 g of $\text{Cu}(\text{NO}_3)_2 \cdot 3\text{H}_2\text{O}$ prepared in 20 ml aqueous was mixed with the above solutions drop by drop under a magnetic stirrer for 20 mins. Further, $\text{Co}(\text{NO}_3)_2 \cdot 6\text{H}_2\text{O}$ of preferred molar ratios like 0.01, 0.02, 0.03, and 0.04 M prepared in 20 ml aqueous were mixed drop by drop. Finally, 0.2M of NaOH pellets were added and poured into the above solution till the pH value reached ~ 9 . The entire solution was continuously agitated for 2 hrs and heated at 80°C . To remove contaminants, the precipitate was washed multiple times with distilled water and ethanol after being filtered. The obtained final product was dried in hot air oven for 6 hours at 60°C and

samples were subsequently calcinated at 450°C for 7 hrs. The annealed powders were pulverized into fine particles using an agate mortar for further characterization.

3. Characterization techniques

The structural properties of prepared pure and dual doped \square -MnO₂ nanorods was studied by X-ray diffraction technique using Philips powder diffractometer (1729 PW) equipped with a monochromatic CuK α radiation source ($\lambda=1.5406\text{\AA}$) for 2θ varying from 20°-60°. The FTIR spectroscopy with the Thermo Nicolet 380 equipment was used to investigate the vibrational bands of synthesized materials.

Results and discussion

3.1. Structural analysis

The structural analysis of pure and \square -MnO₂(_{0.95-x}) Cu²⁺(_{0.05}) Co(_x) nanorods with different concentration levels of cobalt doping ($x= 0.01, 0.02, 0.03$ and 0.04 M) are displayed in Fig. 1. The observed diffraction peaks are widened and high intensity suggested that prepared samples are well crystalline in nature. Also, the XRD spectra of all diffraction peaks were well accordance with the tetragonal crystal structure (JCPDS card no. 41-0141). The pure sample showed the diffraction peaks were related to the characteristic peaks of α -MnO₂ nanorods [26]. The characteristic peaks intensity was increased with increasing Co concentration upto 0.03M. The observed diffraction peaks located at 18.08°, 28.74°, 36.58°, 37.72°, and 49.90°, corresponding to the (220), (310), (400), (211) and (411) planes, respectively. These results indicated that the Co doped samples did not alter the crystal structure of \square -MnO₂:Cu nanoparticles at lower concentrations. The lack of secondary peaks demonstrated the successfully incorporation of Co²⁺ ions into the host lattice, where they occupied Mn²⁺ without disrupting the tetragonal structure. Additionally, no characteristic peaks of cobalt oxide or cobalt hydroxide were observed up to 0.03M Co incorporation, which revealed that the dopant is well integrated into the lattice site during the preparation process. Further, the Co concentration was slightly raised to 0.04M, and the diffraction peaks intensity decreased.

The Debye-Scherrer's formula was used to evaluate the crystallite size of prepared samples [27].

$$D = \frac{k\lambda}{\beta \cos\theta} \quad (1)$$

where, λ was wavelength of the X-ray ($\lambda=1.5406 \text{\AA}$), θ was the angle of the diffraction, k was shape factor, and β was full width half maxima of the peak. According from the Table. 1 indicated that the crystallite size of the dual doped \square -MnO₂ nanorods decreased from 18 to 10 nm with increasing Co concentrations. The decreased crystallite size with increasing Co concentrations due to the observed diffraction peaks was broader. Also, the lattice constant was significantly raised when dual doped into the \square -MnO₂ crystal structure, which results in lattice distortion and a further decreased crystallite size.

The lattice constant of the synthesized samples were estimated using the equation [28],

$$a = \frac{\lambda \sqrt{h^2 + k^2 + l^2}}{2 \sin\theta} \quad (2)$$

where, hkl is the miller indices of the crystal and d is interplanar spacing. The estimated values for lattice constant were observed to be in the range of (a) 9.772–9.787 and (c) 2.853-2.866 \AA for pure and dual doped \square -MnO₂ nanorods, respectively. A small variation was observed in the lattice constant values with an increase in the Co²⁺ ratio, and it was in reasonable agreement with the JCPDS values.

The dislocation density and microstrain of pure and dual doped \square -MnO₂ samples were determined by following the formulae [29,30],

$$\delta = \frac{1}{D^2} \tag{3}$$

$$(\epsilon) = \frac{\beta}{4 \tan \theta} \tag{4}$$

The dislocation density values increased with raising Co content attributed to decreased crystallite size. The calculated strain values increased with increased Co ratio due to the radius of Co^{2+} ions is significantly higher than that of Mn^{2+} ions, resulting in system instability.

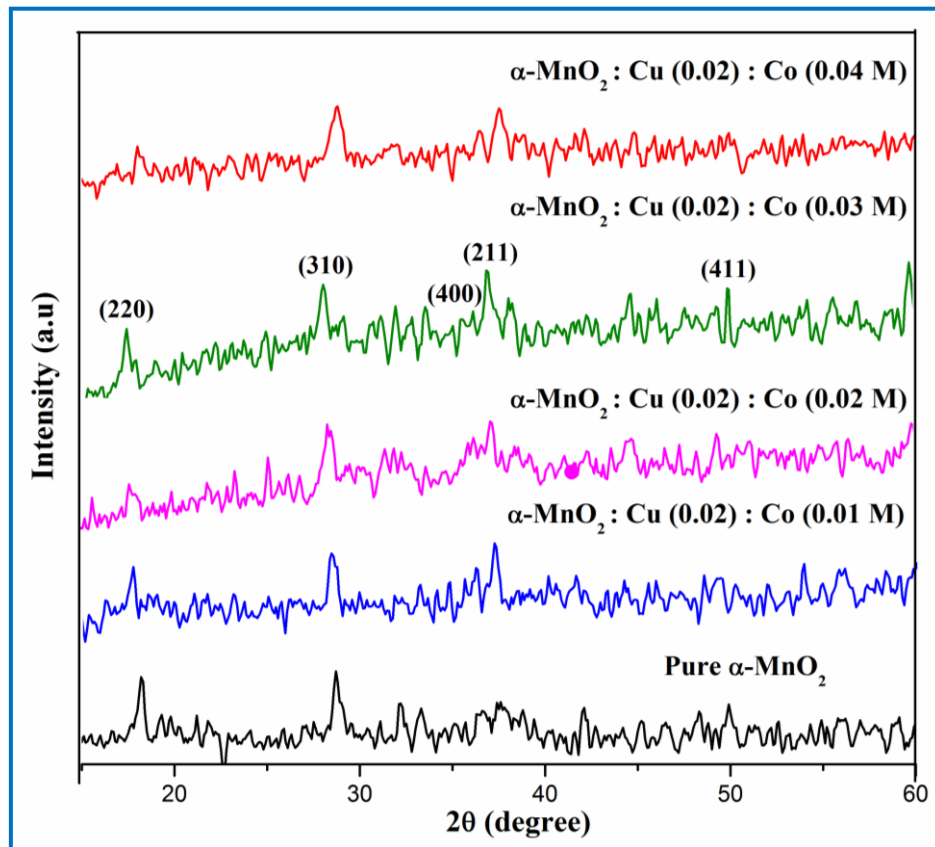


Fig. 1. XRD pattern of pure and dual doped α - MnO_2 nanorods.

Table 1. The structural parameters for dual Cu-Co doped α - MnO_2 nanoparticles.

Samples	Crystallite size (D) nm	Dislocation density (δ) $\times 10^{15}$ (lines/m ²)	Micro strain (ϵ)	Lattice constant (a) Å	
				a	c
Pure α - MnO_2	18	3.0864	0.5087	9.772	2.853
α - MnO_2 : Cu(0.02M) : Co(0.01M)	15	4.4450	0.7906	9.776	2.855
α - MnO_2 : Cu(0.02M) : Co(0.02M)	13	5.9172	0.8214	9.781	2.860

α -MnO ₂ : Cu(0.02M) : Co(0.03M)	12	6.9445	0.8472	9.784	2.865
α -MnO ₂ : Cu(0.02M) : Co(0.04M)	10	10.0012	1.0245	9.787	2.866

3.2. FTIR analysis

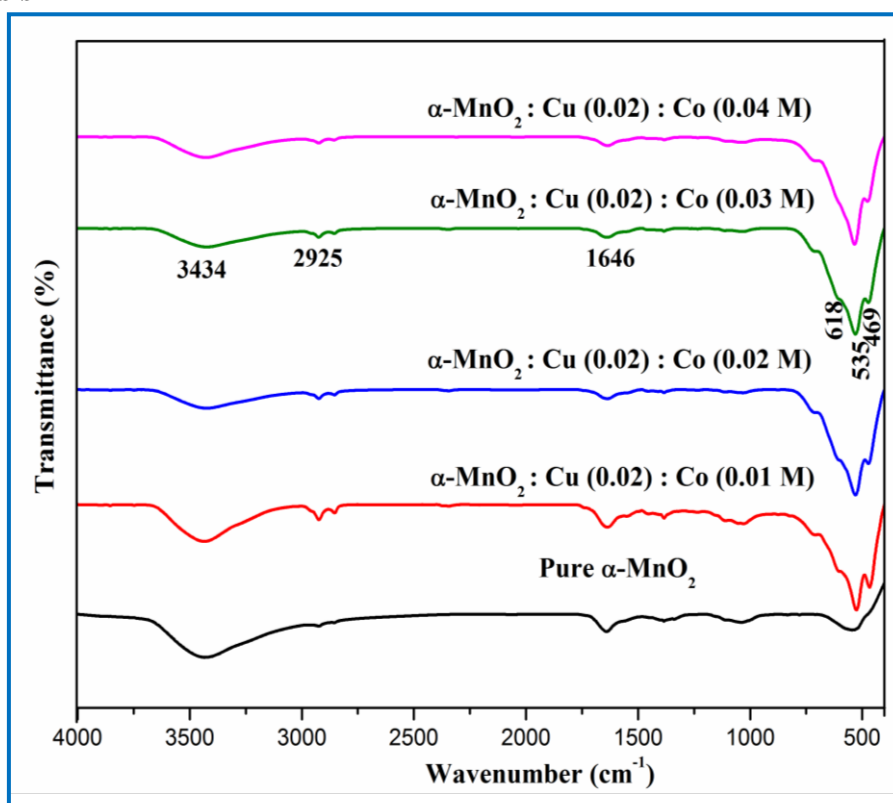


Fig. 2. FTIR spectra of pure and dual doped α -MnO₂ nanorods.

The molecular vibration characteristics of the α -MnO₂ nanorods with varying structures were investigated by recording FT-IR spectra. Fig. 2 displays the absorption bands measured in the range of 400–800 cm⁻¹ were caused by the vibrations of Mn–O and Mn–O–Mn, which were clearly observed in the spectra of all four samples. The stretching vibration of the H₂O molecule and OH⁻ in the lattice was responsible for the broad peak at 3434 cm⁻¹. The Co²⁺ and Cu²⁺ dual doped α -MnO₂ nanorods exhibited the strongest stretching vibration, while the OH⁻/H₂O stretching peak of the α -MnO₂ nanorods was nearly impossible to distinguish [31,32]. The phenomena can be attributed to the direct binding of hydroxyl groups and interlayer hydrates to the intercalated metal ions in the interlayer. The absorption band detected at 1646 cm⁻¹ which is due to bending vibration. Also, the small absorption band detected at 2925 cm⁻¹ can be attributed to the C-H stretching mode [33]. Compared to pure α -MnO₂ sample, dual doped samples intensity decreases along with characteristic peaks were shifted in lower angel side. The aforementioned phenomenon indicates that the phase of α -MnO₂ varies with metal intercalation, and the structure change was activated in a different content or manner by Co²⁺ and Cu²⁺ due to their varying valence and doping amounts. The XRD results were further confirmed by the FTIR results, which showed that no impurity phase was observed.

4. Conclusion

α -MnO₂ nanorods. Co-Cu-doped MnO₂ nanorods were produced using chemical precipitation and examined for structural and functional groups. Structural investigation revealed the tetragonal structure of α -MnO₂, supporting XRD findings at all Co concentrations. We successfully developed an electrode material for supercapacitor applications that utilizes pure and cobalt-copper dual doped Nanorod-like structures were seen in both pure and dual-doped α -MnO₂ samples. The length/diameter of the nanorods decreased as Co concentration increased.

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